

**Supplementary Information for  
Enhanced Anomalous Hall Effect of Pt on an Antiferromagnetic Insulator with  
Fully Compensated Surface**

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The field dependence and the temperature dependence of Hall resistance  $R_H$  and anomalous Hall resistance  $R_{AHE}$  of Pt/YIG and Pt/LSMO are compared side by side in Fig. S1. The sign reversal around 50 K for the anomalous Hall resistance  $R_{AHE}$  of the 10 nm Pt/YIG heterostructure is clearly seen in Fig. S1 (b), while  $R_{AHE}$  of the 2 nm Pt/3UC LSMO remains negative at all measured temperatures in Fig. S1 (d). The  $R_{AHE}$  value of the 10 nm Pt/YIG at 300 K is one order of magnitude smaller than the ordinary Hall resistance  $R_{OHE}$ . Without the subtraction of  $R_{OHE}$ , the contribution of  $R_{AHE}$  at 300 K is almost invisible in Fig. S1 (a). In stark contrast, the  $R_{AHE}$  value of the 2 nm Pt/3UC LSMO at 300K is almost 3 times larger than the  $R_{OHE}$  value as shown in Fig. S1 (c) and (d).

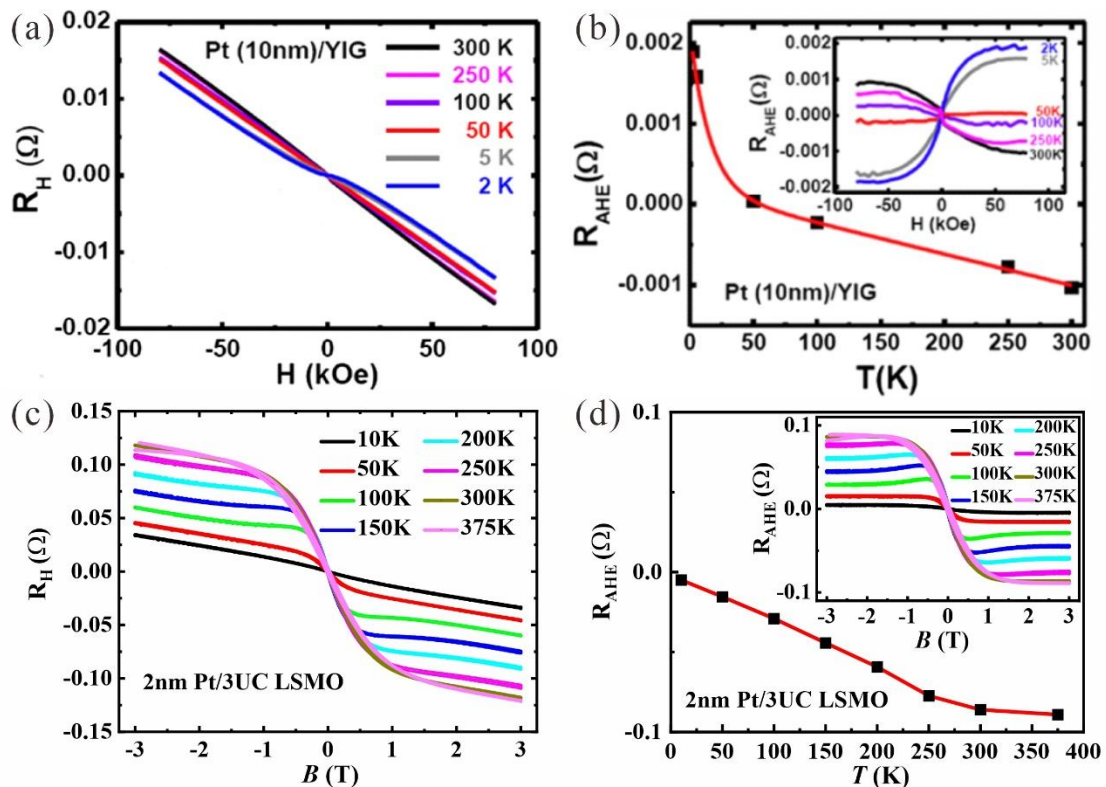


Fig. S1 (a) Field dependence of Hall resistance  $R_H$  at different temperatures for 10 nm Pt/YIG (copied from Ref. [1]). (b) The anomalous Hall resistance  $R_{AHE}$  as a function of the temperature for 10 nm Pt/YIG (copied from Ref. [1]). The inset shows  $R_{AHE}$  as a function of magnetic field. (c) Field dependence of  $R_H$  at different temperatures for

2 nm Pt/3UC LSMO. (d)  $R_{\text{AHE}}$  as a function of the temperature for 2 nm Pt/3UC LSMO. The inset shows  $R_{\text{AHE}}$  as a function of magnetic field.

The diffusion of Mn into the Pt at the Pt/LSMO interface is important, but the diffusion of Mn itself can't yield sizable AHE. As reported by Ning An *et al.* [2], the  $\text{Mn}_{1-x}\text{Pt}_x$  epitaxial films can show large AHE, but only at the composition of  $\text{Mn}_3\text{Pt}$ , which is known as a noncollinear antiferromagnet in  $L1_2$  structure. It is deposited under the optimized sputtering power of Mn target to make sure the right  $\text{Mn}_3\text{Pt}$  stoichiometry and annealed at  $600^\circ\text{C}$  to make sure the right  $L1_2$  structure. These critical conditions don't fit in our case, where the Mn impurities in the Pt are diluted. To know the role of the interdiffusion of Mn, we measured the Hall resistivity of 2 nm  $\text{Pt}_{100}\text{Mn}_1/\text{SrTiO}_3(001)$ . As shown in the Fig. S2 (a), AHE is almost unvisitable. It proves that only the diluted Mn impurities in Pt at the interface are not sufficient to exhibit AHE, and it needs the help of underlying LSMO through the exchange interaction.

In addition, we have measured the Hall resistivity of 2 nm Pt/2 nm Mn/ $\text{SrTiO}_3(001)$ . As shown in the Fig. S2 (b), although  $T_N$  of bulk Mn is 95 K, AHE clearly appears at 100K and gets significantly enhanced at 300 K. Since both Mn and Pt layers are conducting, the contribution of AHE from Mn layer and Pt layer are mixed. However, by comparing the magnitude of  $\rho_{\text{AHE}}$  of 2 nm Pt/2 nm Mn sample ( $1.5 \times 10^{-1} \mu\Omega\cdot\text{cm}$ ) and 2nm Pt/3 UC LSMO sample ( $1.8 \times 10^{-2} \mu\Omega\cdot\text{cm}$ ) at 300K, we can conclude AHE is mainly from the Mn layer due to the short-range order above  $T_N$ .

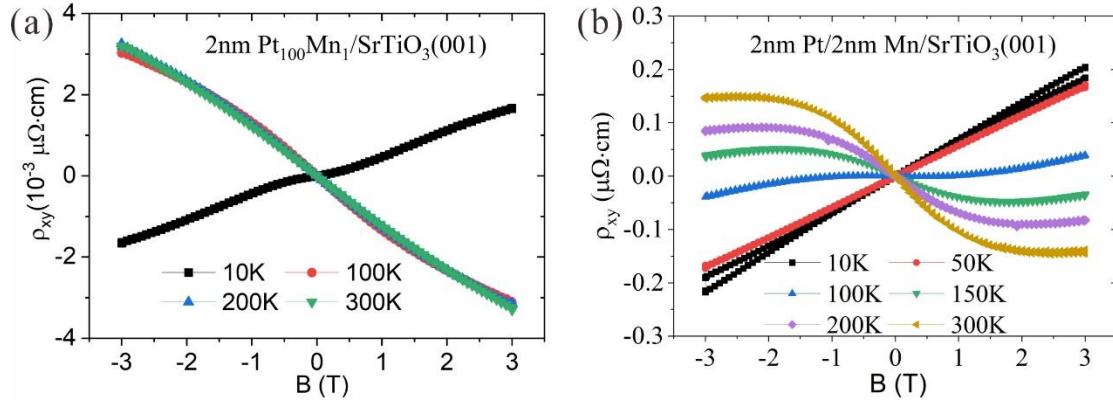


Fig. S2 (a) Field dependence of Hall resistivity at different temperatures for 2 nm  $\text{Pt}_{100}\text{Mn}_1/\text{SrTiO}_3(001)$ . (b) Field dependence of Hall resistivity at different temperatures for 2 nm Pt/2 nm Mn/ $\text{SrTiO}_3(001)$ .

### Reference

- [1] S.Y. Huang, X. Fan, D. Qu, Y.P. Chen, W.G. Wang, J. Wu, T.Y. Chen, J. Q. Xiao, and C.L. Chien, Physical Review Letters 109, 107204 (2012).
- [2] N. An, M. Tang, S. Hu, H.L. Yang, W.J. Fan, S.M. Zhou, and X.P. Qiu, Sci. China-Phys. Mech. Astron. 63, 297511 (2020).